

Principles of X-ray scattering by macromolécules

Ecole RéNaFoBiS, Oléron, juin 2014



1ère Ecole RéNaFoBiS, 1 - 7 juin 2014, Oléron, D. Housset

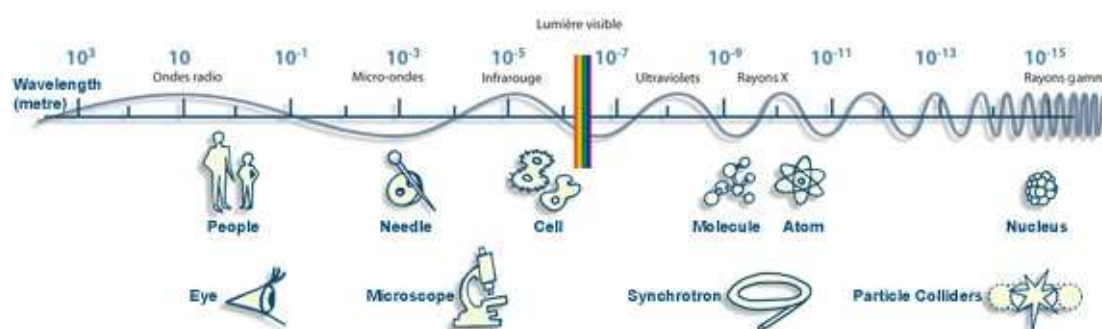
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Parcours



- Physicien (DEA en physique fondamentale en 1987)
- Thèse sur la dynamique moléculaire par simulation sur une protéine, l'utéroglobine (1990)
- Post-doc (Service Militaire) au National Cancer Institute (Maryland): cristallographie sur le BPTI et une asparaginase utilisée en cancérothérapie
- Chercheur CEA à l'Institut de Biologie structurale
 - Cristallographie sur différentes protéines
 - Toxine de scorpion et de serpent => résolution atomique
 - Deshydrogénase
 - Ferredoxine
 - GTPases
 - Récepteur pour l'antigène du lymphocyte T
 - Immunologie structurale
 - Interaction, reconnaissance, lien entre interaction à l'échelle moléculaire et activation au niveau cellulaire

- Based on interaction between radiations and matter
- The wavelength of the radiation used should be suited to the size of the smallest details we want to observe:
 - Resolution limit in diffraction experiments is $\lambda/2$
- Proteins and macromolecules assemblies are too small (1 to 50 nm, 10 to 500 Å) to be observed with visible light ($300 \text{ nm} < \lambda < 800 \text{ nm}$)
- To see atoms, a wavelength of about 1 Å is required



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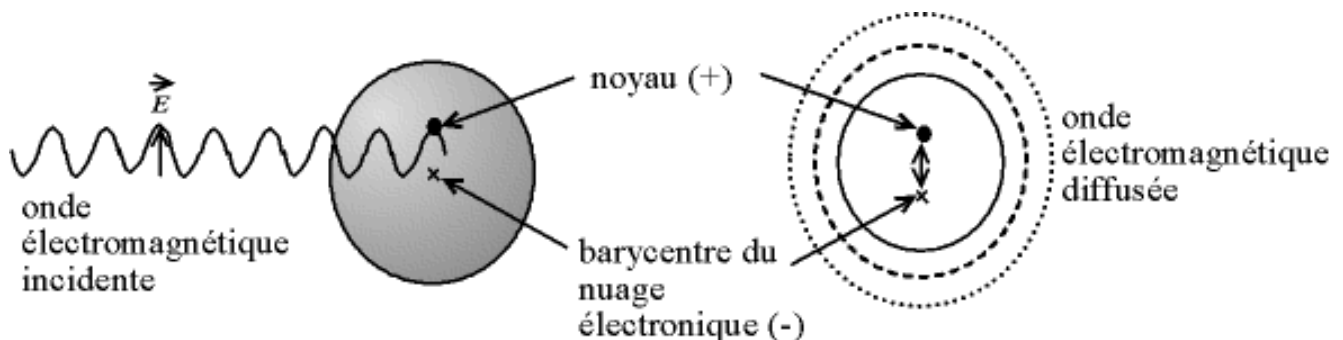
Three types of “radiation” commonly used

- X-rays
 - Electromagnetic waves / photons
 - Photon energy $w = hc/\lambda = h\nu$ with $h = 6.6257 \cdot 10^{-34} \text{ J.s}$ (Plank constant)
 - $7 \text{ keV} < w < 17 \text{ keV}$ or $1.7 \text{ Å} > \lambda > 0.7 \text{ Å}$
- Neutrons
 - Neutral particle ($m_n = 1.6749 \cdot 10^{-27} \text{ kg}$)
 - Energy $w = 1/2 m_n v^2$, wavelength $\lambda = h / m_n v$
 - $\lambda = 1.5 \text{ Å}$ for $v = 2600 \text{ m/s}$ and $w = 3.6 \cdot 10^{-2} \text{ eV}$
- Electrons
 - Negatively charged particle ($q = 1.6 \cdot 10^{-19} \text{ C}$, $m_e = 9.1091 \cdot 10^{-31} \text{ kg}$)
 - Energy $w = 1/2 m_e v^2$, wavelength $\lambda = h / m_e v$
 - $\lambda = 1.2 \text{ Å}$ for $v = 6000 \text{ km/s}$ and $w = 100 \text{ eV}$
- Here, we will focus only on X-rays!

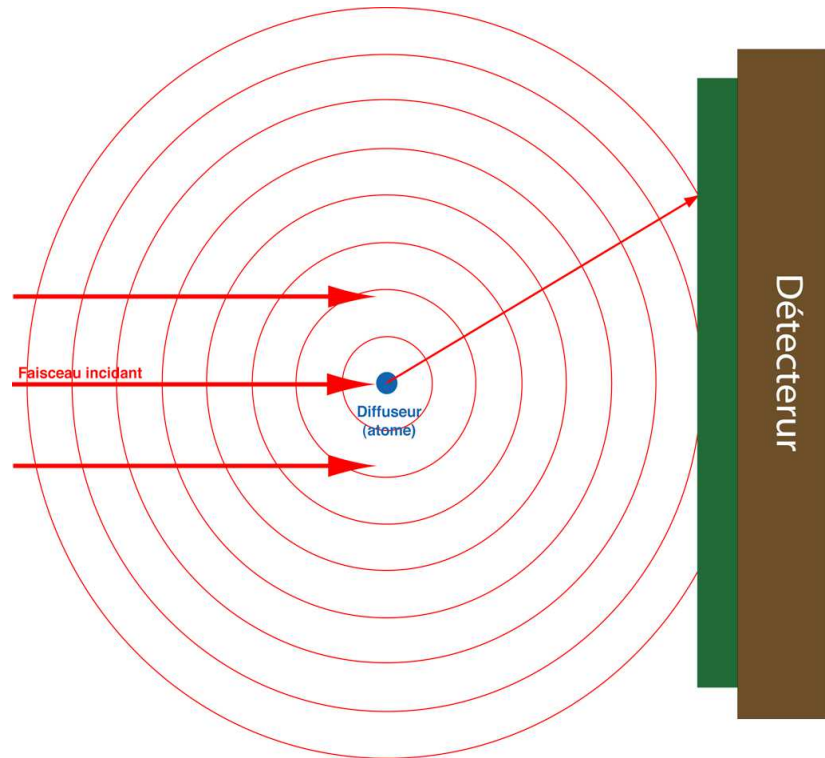
- Elastic scattering (no loss of energy, wavelength is conserved)
 - Rayleigh scattering: bound electron (photon energy $\ll$ electron binding energy)
 - Thomson scattering: free electrons (photon energy $\gg$ electron binding energy)
 - Carbon atom $E(1s) = -1013$ eV, $E(2s,2p) = -36$ eV to be compared to 7 to 15 keV for X-ray photons
 - Photon energy should differ from element absorption edges
- Inelastic scattering (loss of energy, scattered photon have a larger wavelength)
 - Compton scattering (a bound electron is ejected in the continuum)
 - Raman effect (a bound electron jump to a free state of the atom or molecule)
- Ionisation

How a photon is scattered by an atom ?

- Classical model
 - The photon is scattered by the electronic cloud of the atom
 - The incident electromagnetic wave induce the oscillation of the electronic cloud
 - The negatively charged electronic cloud et the positively charged nucleus become an oscillating dipole, thus **emitting a spherical electromagnetic wave of same wavelength and a phase difference of π .**

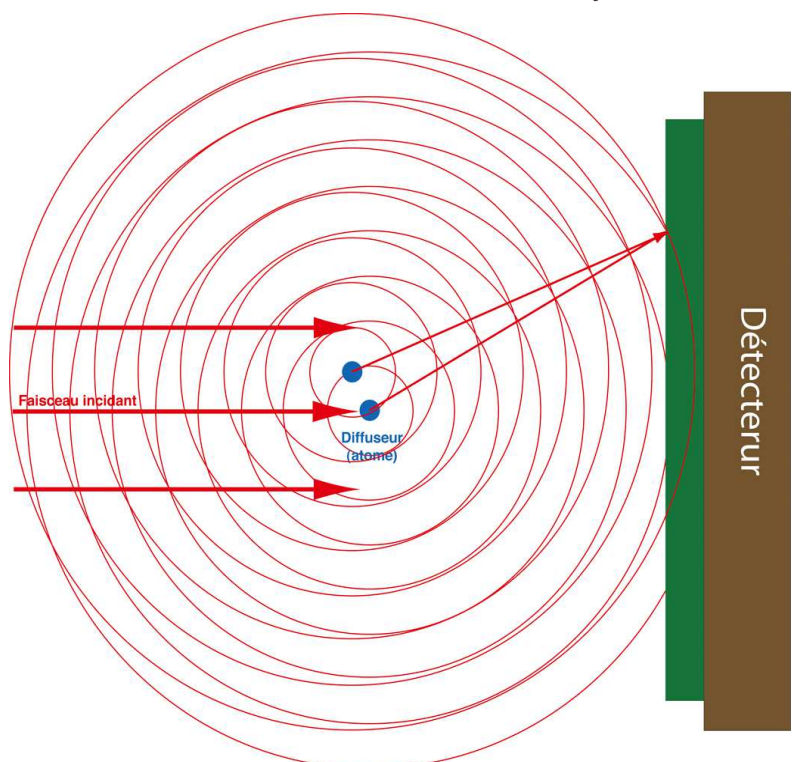


- The planar incident wave induces the emission of a spherical wave of same wavelength and with a π phase shift



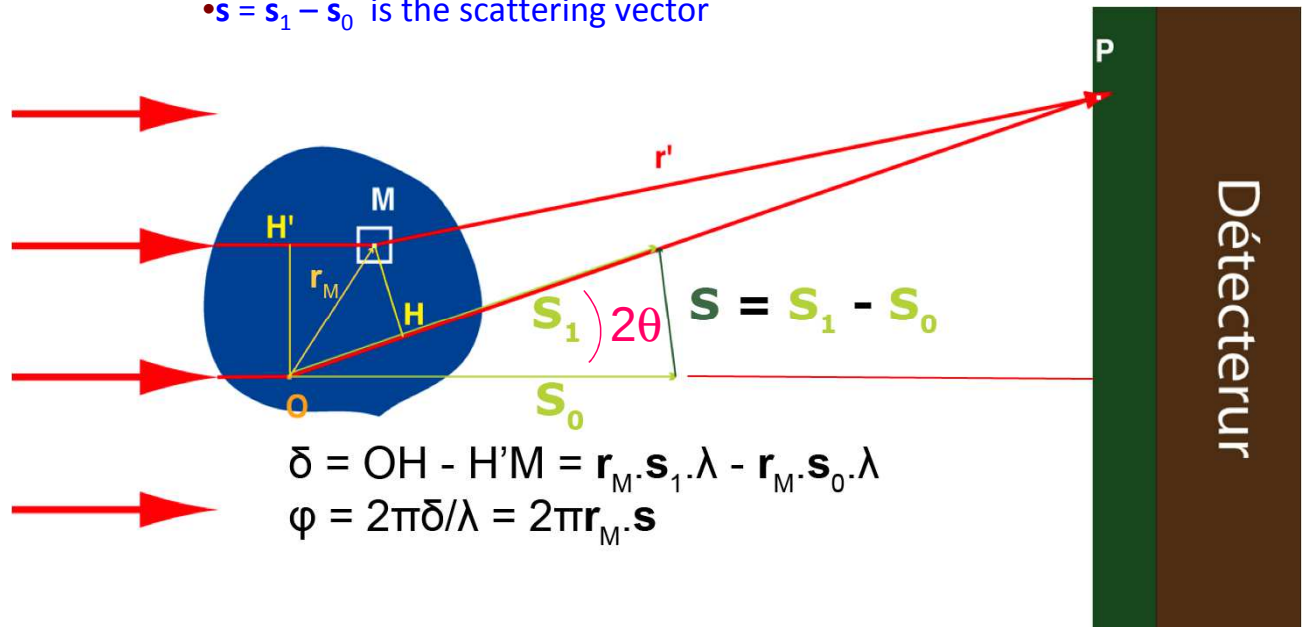
Scattering by two atoms

- The two atoms are emitting a spherical electromagnetic wave. If the two atoms are not mobile and close by, the two waves interfere



What do we measure on the detector ?

- Incident wave described by its wave vector $\mathbf{q}_0$ ($|\mathbf{q}_0| = 2\pi/\lambda$) or $\mathbf{s}_0 = \mathbf{q}_0 / 2\pi$ ($|\mathbf{s}_0| = 1/\lambda$) : $E_0 \cos[\omega t - \mathbf{q}_0 \cdot \mathbf{r}]$
- Scattered wave described by its wave vector $\mathbf{q}_1$ ($|\mathbf{q}_1| = 2\pi/\lambda$) or $\mathbf{s}_1 = \mathbf{q}_1 / 2\pi$ ($|\mathbf{s}_1| = 1/\lambda$)
- $\mathbf{s} = \mathbf{s}_1 - \mathbf{s}_0$ is the scattering vector



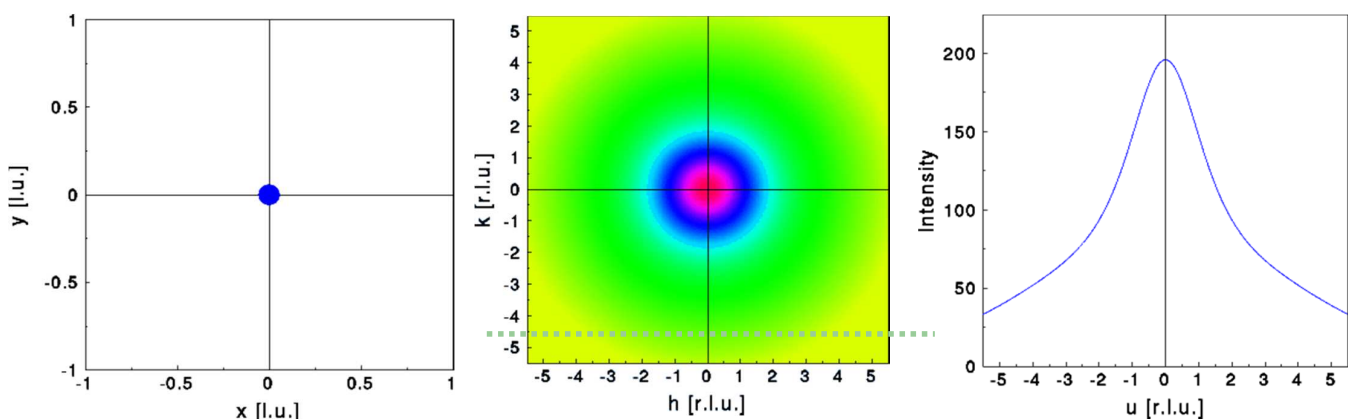
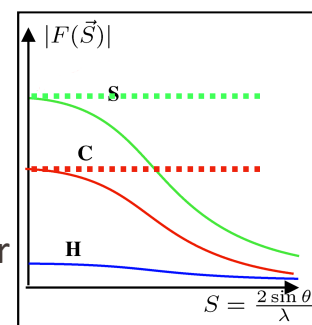
Relation between scattered wave and solid sample?

- Incident wave : $E_0 \exp[i(\omega t - \mathbf{q}_0 \cdot \mathbf{r})]$
- $f(\mathbf{r}_M)$ is the scattering power at point M defined by $\mathbf{r}_M$
 - The wave scattered by the small volume ∂v around M, as observed in P is:
 - $\partial E(\mathbf{s}) = E_0/r' f(\mathbf{r}_M) \exp[i(\omega t - \mathbf{q}_0 \cdot \mathbf{r}_M)] \cdot \exp(-i\mathbf{q}_1 \cdot (\mathbf{r} - \mathbf{r}_M)) \cdot \exp(i\pi) \partial \mathbf{r}_M$
 - $= E_0/r' f(\mathbf{r}_M) \exp(i(\omega t - \mathbf{q}_1 \cdot \mathbf{r})) \cdot \exp(2i\pi \mathbf{r}_M \cdot \mathbf{s}) \cdot \exp(i\pi) \partial \mathbf{r}_M$
- To calculate the wave scattered by the entire sample, we have to sum on the entire volume of the sample:
 - $E(\mathbf{s}) = \int_V E_0/r' f(\mathbf{r}_M) \exp(i(\omega t - \mathbf{q}_1 \cdot \mathbf{r})) \cdot \exp(2i\pi \mathbf{r}_M \cdot \mathbf{s}) \cdot \exp(i\pi) \partial \mathbf{r}_M$
 - $= E_0/r' \exp(i(\omega t - \mathbf{q}_1 \cdot \mathbf{r})) \exp(i\pi) \int_V f(\mathbf{r}_M) \exp(2i\pi \mathbf{r}_M \cdot \mathbf{s}) \partial \mathbf{r}_M$
 - $E(\mathbf{s})$ is proportional to the Fourier transform of the scattering power of the sample
- For X-rays, this scattering power is proportional to the electron density
 - $E(\mathbf{s})$ is proportional to the Fourier transform of the electron density

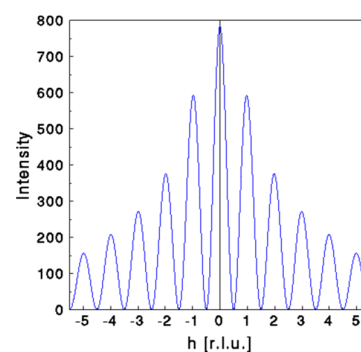
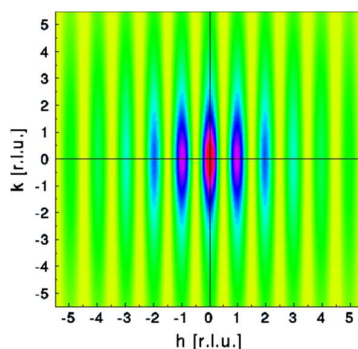
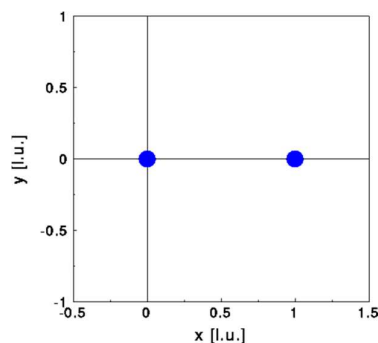
- $E(\mathbf{s}) = E_0/r' \sigma \int_V \rho(\mathbf{r}) \exp(2i\pi \mathbf{r} \cdot \mathbf{s}) d\mathbf{r}$
- Structure factor:
 - Direct Fourier transform of the electron density
 - $F(\mathbf{s}) = \int_V \rho(\mathbf{r}) \exp(2i\pi \mathbf{r} \cdot \mathbf{s}) d\mathbf{r}$
 - Complex number (amplitude and phase)
- Electron density
 - Indirect Fourier transform of the structure factors
 - $\rho(\mathbf{r}) = \int_V F(\mathbf{s}) \exp(-2i\pi \mathbf{r} \cdot \mathbf{s}) d\mathbf{s}$
- The detector measures the intensity of the scattered wave
- The intensity of the scattered wave is proportional to the square of the structure factor modulus
 - $I(\mathbf{s}) \propto F(\mathbf{s}) \cdot F^*(\mathbf{s}) = |F(\mathbf{s})|^2$

A few exemples ...

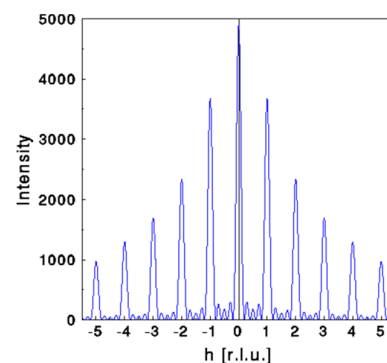
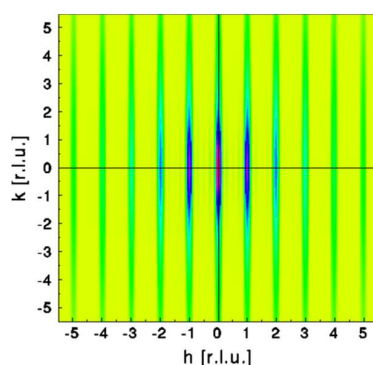
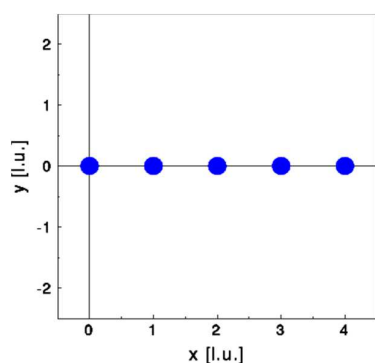
- Diffusion spectra of one atom
 - Spherical electron density
 - $f_{\text{at}}(s) = 2 \int_{r=0,+\infty} \rho_{\text{at}}(r) r \sin(2\pi sr) / s dr$
 - $f_{\text{at}}(0)$ = number of electrons in atom
 - $f_{\text{at}}(s)$ is constant when atom is ponctual (neutron diffraction)
 - $f_{\text{at}}(s)$ is the atomic scattering factor



■ Diffusion by two atoms



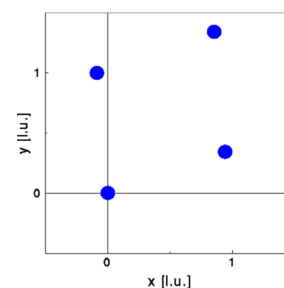
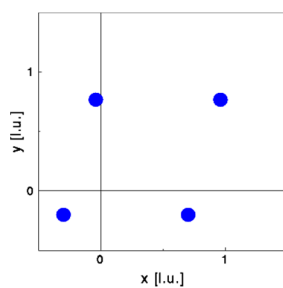
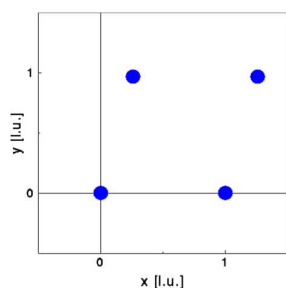
■ Diffusion by five atoms



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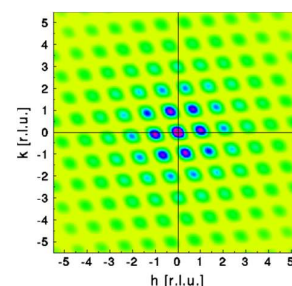
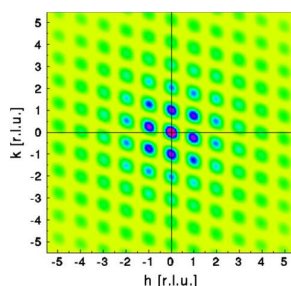
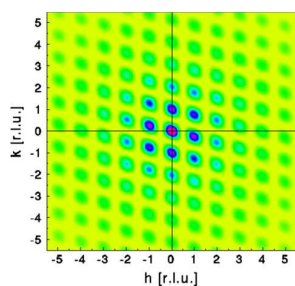
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■ Four atoms: effect of translation and rotation of the set of atoms



➤ No change upon translation

➤ Diffusion spectra is rotated when the set of atoms are rotated



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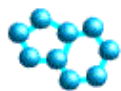
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- $f_j(s)$ is the scattering factor of the atom j
- The phase shift of the wave scattered by an atom in position $\mathbf{r}_j$, in the direction defined by scattering vector $\mathbf{s}$ is (relatively to an atom at the origin):
 - $2\pi\mathbf{r}_j\mathbf{s}$
- The structure for an assembly a N atoms is thus:
 - $F(\mathbf{s}) = \sum_{j=1,N} f_j(s) \cdot \exp(2i\pi\mathbf{r}_j\mathbf{s})$
- $f_j(s)$ are known and tabulated for each element
 - Calculating the structure factor of an assembly of atoms is straightforward

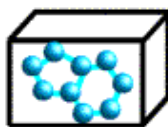
A specific case of solid sample: the crystal

- The crystal is described by a lattice which characterize the three-dimensional periodicity
- The lattice is define by three vector, **a**, **b**, **c**, that define the unit cell
- The content of each unit cell is identical

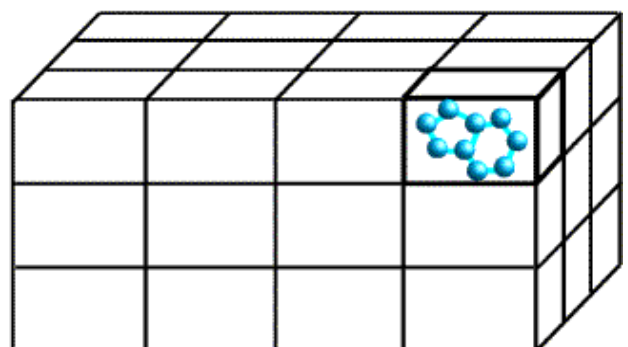
molecule



unit cell



crystal



Structure factor of a crystal

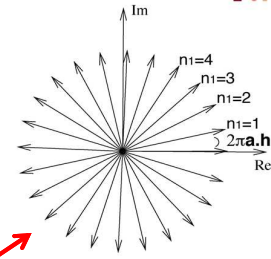
- The structure factor is:

- $F(\mathbf{s}) = \int_{\text{vech.}} \rho(\mathbf{r}) \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r}$

- $F(\mathbf{s}) = \sum_n \int_{\text{vmaille}} \rho(\mathbf{r} + \mathbf{r}_n) \exp(i 2\pi \mathbf{s} \cdot (\mathbf{r} + \mathbf{r}_n)) d\mathbf{r}$

- $F(\mathbf{s}) = \sum_n \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}_n) \int_{\text{vmaille}} \rho(\mathbf{r}) \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r}$

- $F(\mathbf{s}) = \sum_n \exp(2i\pi n_1 \mathbf{a} \cdot \mathbf{s}) \exp(2i\pi n_2 \mathbf{b} \cdot \mathbf{s}) \exp(2i\pi n_3 \mathbf{c} \cdot \mathbf{s}) \int_{\text{vmaille}} \rho(\mathbf{r}) \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r}$



- This structure factor close to zero, except for specific values of $\mathbf{s}$:

- $\mathbf{a} \cdot \mathbf{s} = h, \mathbf{b} \cdot \mathbf{s} = k, \mathbf{c} \cdot \mathbf{s} = l$ where h, k, l are integers (Laue equations)

- Diffraction spots instead of diffusion spectra

- Discrete structure factors

- In directions that satisfy Laue equation, the structure factor is the one of the unit cell, multiplied by the number of cells in the crystal.

- The crystal is a signal amplifier!

Reciprocal lattice

- If $\mathbf{s}$ satisfy the Laue equation:

- $\mathbf{a} \cdot \mathbf{s} = h, \mathbf{b} \cdot \mathbf{s} = k, \mathbf{c} \cdot \mathbf{s} = l$ where h, k, l are integers

- Then $\mathbf{s}$ is a vector of a lattice named reciprocal lattice

- $\mathbf{s} = h \cdot \mathbf{a}^* + k \cdot \mathbf{b}^* + l \cdot \mathbf{c}^*$ where

- $\mathbf{a}^* = \mathbf{b} \wedge \mathbf{c} / V_{\text{maille}}, \mathbf{b}^* = \mathbf{c} \wedge \mathbf{a} / V_{\text{maille}}, \mathbf{c}^* = \mathbf{a} \wedge \mathbf{b} / V_{\text{maille}}$

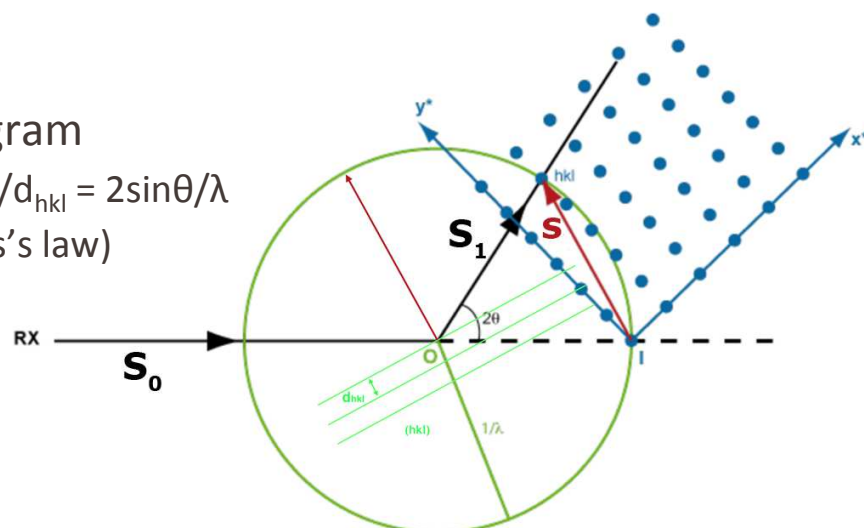
- $V_{\text{maille}} = (\mathbf{a} \wedge \mathbf{b}) \cdot \mathbf{c}$

- Larger unit cell => smaller reciprocal cell => closer spots on diffraction spectra

- Ewald diagram

- $|\mathbf{s}| = 1/d_{hkl} = 2\sin\theta/\lambda$

- (Bragg's law)

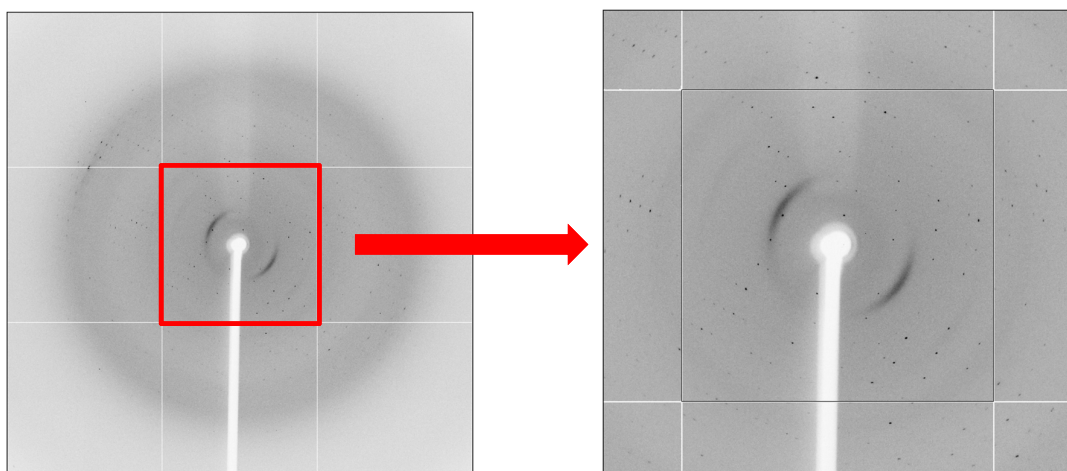


- A parallel with sound
 - in crystals, $\rho(\mathbf{r})$ is a periodic function in the three dimensional space
 - The sound signal is periodic function of time (one dimension)
- $F(\mathbf{s})$ are the coefficient of the development of $\rho(\mathbf{r})$ in a Fourier serie :
 - $F(\mathbf{s}) = \int_{V_{\text{maille}}} \rho(\mathbf{r}) \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r}$
 - $\rho(\mathbf{r}) = 1/V_{\text{maille}} \sum_{\mathbf{s}} F(\mathbf{s}) \exp(-i 2\pi \mathbf{s} \cdot \mathbf{r})$
- The Fourier transform of the sound signal provide the frequencies contained in the sound
 - Include the fundamental frequency and the harmonics (2 times, 3 times, ... the fundamental frequency)
- The largest s or frequency is, the more detailed is the drescription of $\rho(\mathbf{r})$ or of the sound signal

What does the diffraction tell us ?

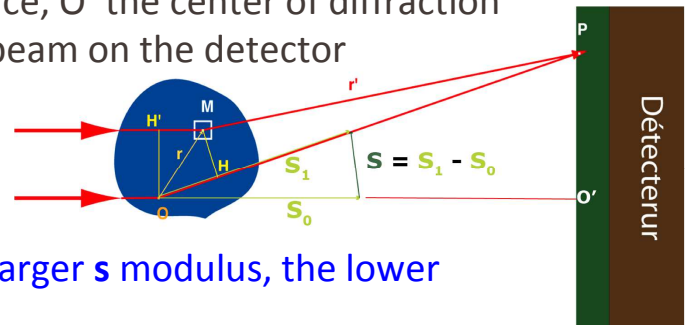
- The **position of spots** on diffraction images will allow the determination of the unit cell dimension and the orientation of the crystal
- The **intensity of each spot** contains information on the electron density of the entire cell
 - $I(\mathbf{s}) = F(\mathbf{s}) \cdot F^*(\mathbf{s})$
 - $= \left(\int_{V_{\text{maille}}} \rho(\mathbf{r}) \exp(i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r} \right) \left(\int_{V_{\text{maille}}} \rho(\mathbf{r}) \exp(-i 2\pi \mathbf{s} \cdot \mathbf{r}) d\mathbf{r} \right)$

Rap1 diffraction image
Courtesy of Dr. Marie-
Hélène Le Du



- In crystallography
 - $d = 1 / |\mathbf{s}|$ in Å
- Directly related with the distance from the center (direct beam position) of the diffraction image:

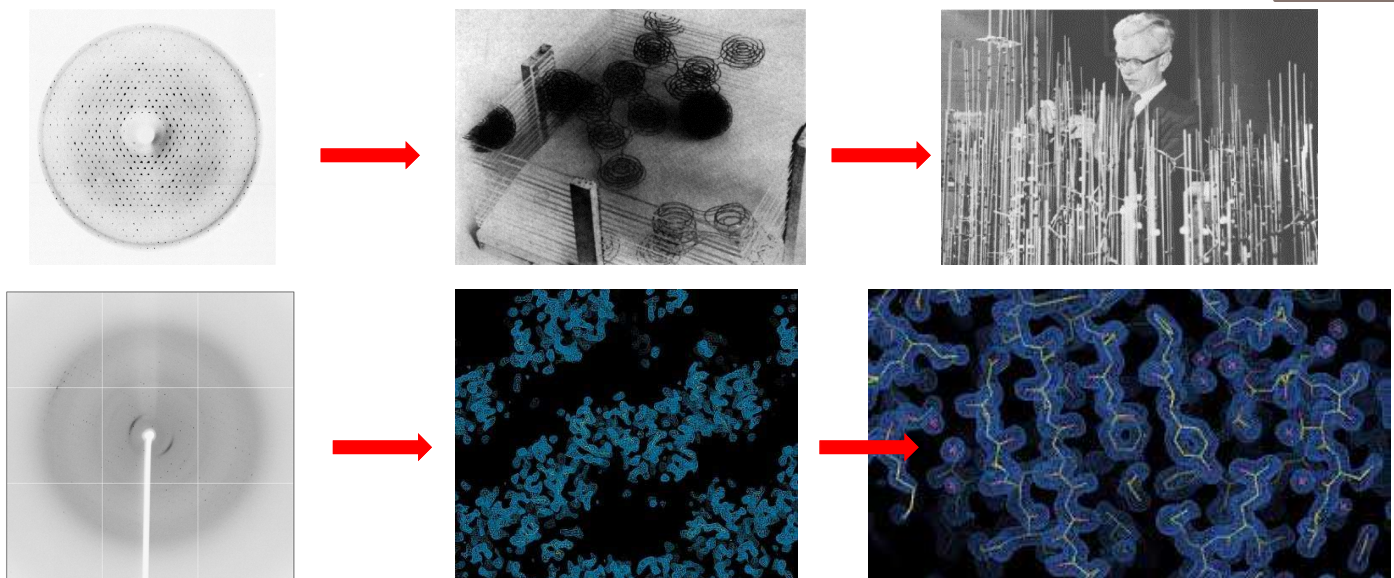
- If D is the crystal - detector distance, O' the center of diffraction image, P the position of diffracted beam on the detector
- $|O'P| = D \sin 2\theta$
- $1 / d = |\mathbf{s}| = 2 \sin \theta / \lambda$



- The further from the center, the larger s modulus, the lower value of d (higher resolution) in Å

- Maximum resolution in diffraction experiments
 - $2\theta = 180^\circ \Rightarrow \theta = 90^\circ \Rightarrow d = \lambda / 2$
- Details that can be seen (separated)
 - $d/\sqrt{2}$ in crystallography. What about SAXS ?

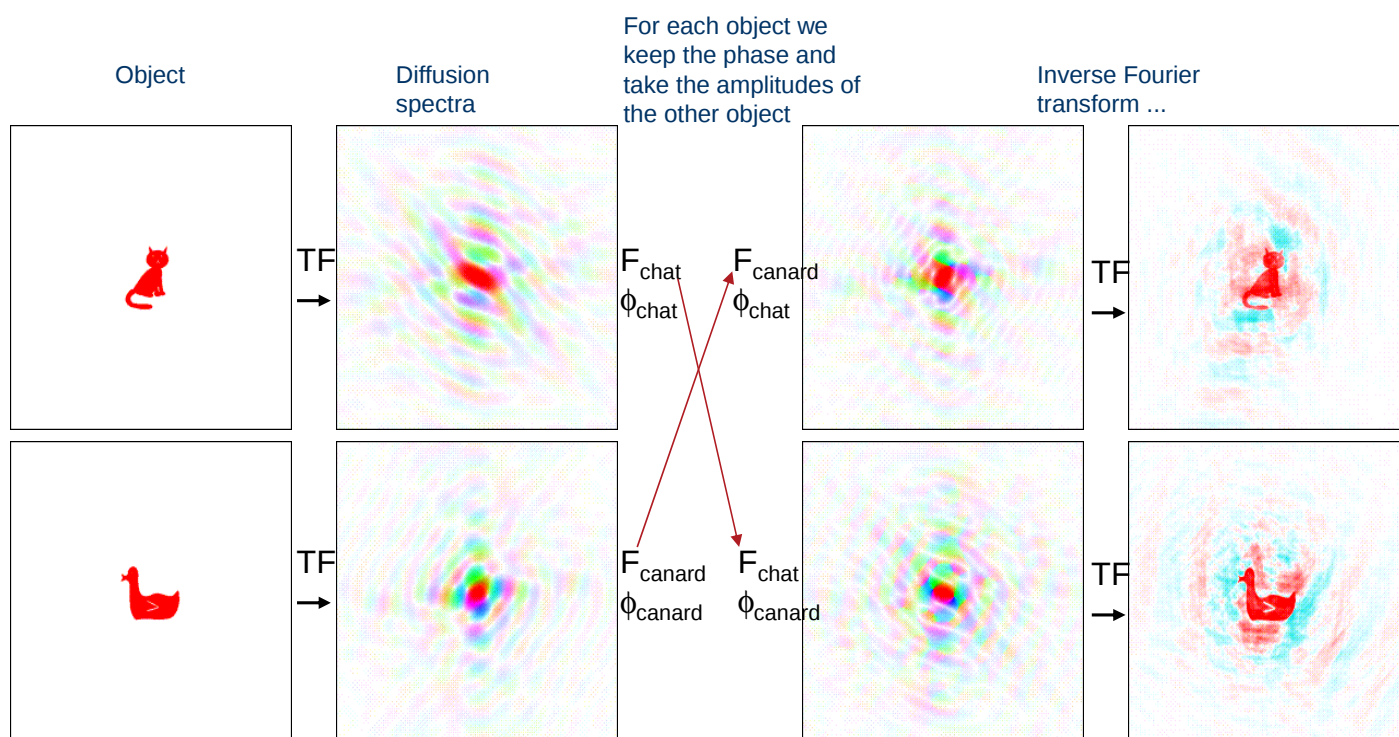
Calculating the electron density ?



- Simple in principle
 - The electron density within the unit cell can be derived from the structure factors :
 - $\rho(\mathbf{r}) = 1/V_{\text{maille}} \sum_{\mathbf{s}} F(\mathbf{s}) \exp(-i 2\pi \mathbf{s} \cdot \mathbf{r})$
 - But ...

- We do not measure $F(\mathbf{s})$ but only an intensity $I(\mathbf{s})$ which is proportional to the square of $F(\mathbf{s})$ modulus: **the phase of the scattered wave is lost**
 - $F(\mathbf{s}) = |F(\mathbf{s})| \exp(i\phi(\mathbf{s}))$
 - Intensity measurements do not allow the calculation of the electron density for macromolecules
- Methods have been developed to provide a reasonable estimate of the phases $\phi(\mathbf{s})$ and allow the calculation of $\rho(\mathbf{r}) \Rightarrow$ Jean-Luc's Talk

Most of the structural information is in the phases

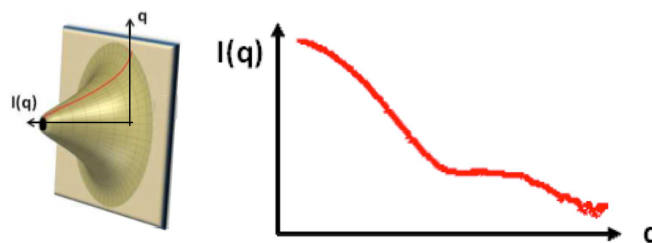


- The Patterson function can be calculated without phase information
 - $P(\mathbf{u}) = 1/V \sum_{\mathbf{s}} F(\mathbf{s}) F^*(\mathbf{s}) \exp(-2i\pi \mathbf{u} \cdot \mathbf{s})$
 - $= 1/V \sum_{\mathbf{s}} |F(\mathbf{s})|^2 \exp(-2i\pi \mathbf{u} \cdot \mathbf{s})$
 - It is the Fourier transform of the squared amplitude of structure factors
- It represents the autocorrelation function of the electron density:
 - $P(\mathbf{u}) = \int_V \rho(\mathbf{r}) \cdot \rho(\mathbf{r}+\mathbf{u}) d\mathbf{r}$
 - $P(\mathbf{u})$ has a local maximum when $\mathbf{u}$ is an interatomic vector
- Can be used to solved structures when the number of atoms is small (less than 50) : not suitable for proteins
- Very useful for phasing (MR, Heavy atom derivatives, anomalous signal)

What about macromolecules in solution ?

- Macromolecules move due to thermal motion
 - Each macromolecules adopts all orientations during the measurements
 - Spherical averaging
- The distance between scatterers is not fixed
 - The distance between two equivalent atoms in two distant molecules vary during the measurement
 - => incoherent scattering
 - => no interference ($\neq$ diffraction by a crystal)
 - => Intensities scattered by each molecules adds up
- Scattering of the solvent is subtracted
 - $I_{\text{macromolecule}}(\mathbf{q}) = I_{\text{sample}}(\mathbf{q}) - I_{\text{buffer}}(\mathbf{q})$
- Provide information on electron density difference between the sample and the buffer, $\Delta\rho(\mathbf{r})$

- The scattering spectra has a radial symmetry
- One dimensional data due to the spherical averaging



- $I(q) = \langle \text{TF}(\Delta\rho(\mathbf{r})) \cdot \text{TF}(\Delta\rho(-\mathbf{r})) \rangle$
- $I(q) = \langle \text{TF}(\int_{V_{\text{particle}}} \Delta\rho(\mathbf{r}+\mathbf{u}) \cdot \Delta\rho(\mathbf{u}) d\mathbf{u}) \rangle$
- $I(q) = 4\pi \int_0^{\infty} p(r) \cdot \sin(qr) / qr \, dr$
 - $p(r)$ is the distribution of distances between all pairs of points within the particle weighted by the respective electron densities
- $p(r) = r/2\pi \int I(q) \cdot q \cdot \sin(qr) \, dq$
- Some structural characteristics of the macromolecule can be derived from $p(r)$.

Comparison of diffraction and SAXS spectra

