# Micro and nano-focused X-ray beams, 4<sup>th</sup> generation synchrotron sources, X-ray Free Electron Lasers :

current and future opportunities in structural biology.

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# **Outline :**

## 1: X-ray macromolecular crystallography in an equation-free nutshell

2: Conventional micro-crystallography at synchrotron-radiation (SR) sources

3: Serial micro-crystallography (XFELs and SR sources)

4: Serial nano-crystallography (XFEL sources)

5: Coherent X-ray diffraction and direct phasing of structural information

## Why resorting to the use of X-rays?

Because their wavelength commensurates with atomic dimensions

## Size of biological objects studied in structural biology







To "see" an object, we need a light whose wavelength commensurates with the size of the object



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To "see" an object, we need a light whose wavelength commensurates with the size of the object



7.3

X-rays are very well suited to probe biological materials at atomic resolution

## X-ray wavelength and sources generally used in crystallography



Synchrotron • λ = 0.5-2.5 Å (as far as we are concerned...) • brillance<u>: 10<sup>23</sup> photons/(sec.mrad<sup>2</sup>.mm<sup>2</sup>.0.1%BW</u>)



X-ray free eletron lasers (XFELs) • λ = 0.5-2.5 Å (as far as we are concerned...) • brillance<u>: 10<sup>33</sup> photons/(sec.mrad<sup>2</sup>.mm<sup>2</sup>.0.1%BW)</u>

## **General principle of lens-based imaging**



• A point from the object scatters incident light.

• Emitted light is intercepted by a lens, which re-focuses it as a point on the detector, hence providing an optical image.

- Light is an electromagnetic wave: it has an intensity and a phase.
- Refocusing by a lens allows preserving both the intensity and the phase.
- Waves from the incident light can display different phases: phase difference is angular :
  - $\theta$  = 0 : waves are **in phase**
  - $\theta = \pi$  : waves have **opposed phases** (180°)
  - $\theta$  =  $\pi/2$ : waves are in phase **quadrature** (90°)



#### There are no lens that can refocus X-rays better than 5-10 nm

- Emitted waves cannot be refocused :

We therefore need to work in the reciprocal space.

- In diffraction experiment, we only measure intensities and the phase information is lost:

 $\rightarrow$ Computation will be required to get it back

→The mathematical operation that allows going from the real space to the reciprocal space is the **Fourier transform** 

## First things first : what is the reciprocal space ?

Intuitive answer :

an inversion of the real space

Educated answer:

the Fourier transform of the real space

Answer in a crystallography exam :

the scattering space, as opposed to the sample space

## Why even bother about ... the reciprocal space ?

Intuitive answer :

because you have personal issues

Educated answer:

because when it comes to small objects, it makes things easier

(e.g., when you try measuring the thickness of a hair)

Answer in a crystallography exam :

because we don't have X-rays lenses focusing below 10 nm, and therefore cannot refocus X-ray scattering from biological samples

## X-ray scattering from a molecule cannot be refocused



## X-ray scattering needs to be recorded in the reciprocal space



and then...

Sample space

**Scattering space** 

#### X-ray scattering is about probing the structure without a lens



Sample space

## **Scattering space**

**Image space** 

#### X-ray scattering from a molecule



# Sample space

## **Scattering space**

<u>Diffraction Pattern</u>: contains all the contrast relevant information at the resolution of  $\lambda/2\sin(\theta)$ 

#### **Restrictions to the use of X-rays**

X-ray/matter interaction is weak (1/10000 photons)

We use crystals as a means to amplify the signal (2.10<sup>14</sup> symmetry-related molecules)

The Bragg law applies : constructive interferences only if  $2^*d^*sin(\theta) = \lambda$ 



transform; no Bragg spots

fringes between spots

#### X-ray scattering from a crystal





#### **Constructive interferences only if :**

 $2^*d^*sin(\theta) = \lambda$  $d/2 = sin(\theta)/\lambda$ - distance vs. angle "reciprocal" relation - fundamental unit is not  $\theta$ , but sin( $\theta$ )/ $\lambda$ 

## X-ray protein crystallography



#### A typical diffraction pattern



• The crystal is characterized by symmetry-operators :

The protein molecules are "**symmetry-related**" within the crystal (real-space)

• This symmetry also applies in the reciprocal space :

Equivalent diffraction spots must be merged

• We can then attribute indices to the spots :

H, K, L

- The intensity of a diffraction spot is proportional to the square of its amplitude : I  $\alpha$  F<sup>2</sup>

• F is a structure factor - it has h, k, l indices in the reciprocal space :  $F_{hkl}$ 

• F<sub>*hkl*</sub> is a complex number of the reciprocal space

Its "Fourier transform" is the density at point *x*, *y*, *z* in the real space :  $\rho_{xyz}$ 

$$\rho_{x,y,z} = 1/V \cdot \Sigma \Sigma \Sigma |\mathsf{F}_{hkl}| \cdot \exp [\mathrm{i}\varphi_{hkl} - 2\pi \mathrm{i}.(hx + ky + lz)]$$



Amplitude = Brightness/Saturation

Phase = Hue/Color

## What is the Fourier transform of an atom or a protein ?



#### What is the Fourier transform of an atom or protein crystal ?



## Why do we need high resolution data ?



Fourier Transform



Reverse Fourier Transform

## Why do we need high resolution data ?



http://ucxray.berkeley.edu/~jamesh/movies/

#### Why don't we always get high resolution data ?



• The crystal isn't perfect but displays intrinsic mosaicity :

The quality of the diffraction data depends on the crystal





Crystal of good quality

Crystal of bad quality

#### Why do we need low resolution data ?



Fourier Transform



Reverse Fourier Transform

## Why do we need low resolution data ?



#### Why do we need 100 % completeness ?



Fourier Transform



Reverse Fourier Transform

## Why do we need 100 % completeness ?



http://ucxray.berkeley.edu/~jamesh/movies/



Amplitude = Brightness/Saturation

Phase = Hue/Color



#### Amplitude = Brightness/Saturation



#### We only measure amplitudes ; the phase information is lost !!!

## Why are phases so important ?



Fourier Transform











Fourier Transform







**Reverse Fourier Transform** 






## Why are phases so important ?



Fourier Transform











Fourier Transform







**Reverse Fourier Transform** 







 $|\mathsf{F}|$  Jerome Karl  $\Phi$  Jerome Karl



 $|\mathsf{F}|$  Herb Hauptman  $\Phi$  Herb Hauptman

#### Why are phases so important ?



 $m = cos(\Delta \phi)$ Δφ = phase error (°) How can we solve the phase problem ?



Perturb the structure and diffraction pattern : use of heavy atoms or anomalous scatterers

Guess the phases : Molecular Replacement

#### How is molecular replacement possible ?



Measured data : only intensities (amplitude), no phase information.

 $(\boldsymbol{A}_{obs})$ 

Something known to be similar.

We can calculate phases and amplitudes via Fourier transform

 $(\mathbf{A}_{calc}, \Phi_{calc})$ 

Reverse Fourier transform using measured intensities and phases from similar structure

 $(\mathbf{A}_{obs}, \Phi_{calc})$ 

#### In practice, ...





Measured data : only intensities (amplitude), no phase information.

 $(\mathbf{A}_{obs})$ 

Something known to be similar.

We can calculate phases and amplitudes via Fourier transform

 $(\mathbf{A}_{calc}, \Phi_{calc})$ 

Reverse Fourier transform using measured intensities and phases from similar structure

 $(A_{obs}, \Phi_{calc})$ 

#### **Structure determination flowchart**



#### **Structure refinement**

To refine your structure in reciprocal space, you need a weighted function taking into account :

• the geometry of your model (1,2) (covalent bonds and angles, van der Waals and electrostatic interactions, etc ...)

 the agreement with the X-ray experimental data (the model which fits the data best)

the likelihood of your model
(the accuracy of such a model)



#### (2) Atomic displacement parameters



#### **Structure refinement**

From the refined model, better phases, therefore a more reliable electron density map, allowing to build a better model :





## The best model is the one which has the highest probability given a set of observations and a certain prior knowledge.

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## **Conventional crystallography : use of oscillation for data collection**



When is it necessary to resort to micro/nano crystallography?

## Heterogeneity in crystal quality

Beam radius: 10 - 2.5  $\mu$ m e.g. crystals of membrane proteins, ribosome, etc.

A









## Small crystals

Beam radius: 1 - 2.5  $\mu$ m e.g. crystals of membrane proteins, ribosome, etc.



## Micro- and nano-crystals

Beam radius: 0.1- 2.5 μm\_

e.g. crystals of membrane proteins, proteins complexes, amyloid proteins



## Micro/nano-beam & micro/nano-focused beams are not the same thing

#### Micro-beam vs. micro-focused beam



# Suited for **small crystals** or for crystals displaying **heterogeneity in diffraction quality**



#### Micro-beam vs. micro-focused beam





#### Micro-beam vs. micro/nano-focused beam





## To collect oscillation data, crystals must be mounted & centered beforehand

#### **Conventional size vs. microcrystals**





**2\*2\*50 μm<sup>3</sup>**, *i.e.* 0.9x10<sup>3</sup> \* 0.9x10<sup>3</sup> \* 100x10<sup>3</sup> = **8.1x10<sup>10</sup> unit cells total** 

10,000 LGNY microcrystals could fit in this lysozyme crystal

How can we mount and centre a microcrystal?

#### Loop vs. mesh



Background scattering from loop
Crystal is hidden in the loop fiber
and is difficult to centre



μ-mesh



- Reduced background scattering
- Easier to see & center crystal
- Improved diffracted intensities

## Loop vs. mesh



KLIMY peptide needles

#### Loop vs. capillaries



Background scattering from loop
Crystal is hidden in the loop fiber
and is difficult to centre

- No background scattering
- Easy to see & center crystal
- Improved diffracted intensities

How to mount a crystal on a glass  $\mu$ -capillary ?

## **Crystal Mounting**



#### **Crystal Mounting**

#### Fish the microcrystals out of the drop using > 10x



30x(....and repeat, and patience.....)

## Crystal Evaluation under microscope



## Screen through dozens of crystals for diffraction quality







#### Can structures be solved from such small crystals?



Colletier, Laganowsky, Zhao, Soriaga, Landau, Goldschmidt, Cascio, Sawaya & Eisenberg (2011). PNAS, 108, 16938-16943

#### Can such small crystals diffract to high resolution ?



## Macromolecular micro-crystallography at ESRF?



## What if crystals are either too small or too fragile,

## and thence unamenable to mounting ?

e.g. membrane proteins, amyloid proteins, protein complexes,

The essential limitation to obtaining well-ordered, large crystals of such objects resides in the limited number of strong and specific contacts that can be established between them inside the crystal.

Thus, they are fragile by nature.

In addition, cryo-protection may tamper with crystal contacts, requiring that data be collected at room-temperature.



#### What if crystals are either too small or too fragile,

## and thence unamenable to mounting ?

e.g. naturally occurring nano-crystals (< 1µm)

#### 1 – Cryo-protection is not an option

- \* issues w/ ice nanocrystallites
- \* issues w/ background scattering from cryo stream

#### 2 – Oscillation data collection is not an option

- \* goniometer's sphere of confusion (1-5  $\mu$ m) >> larger than the crystal
- \* camera resolution ≤ the size of the crystal
- \* optical effects due to solvent & cryo stream
- \* loops & meshes >> than the samples; µ-capillary not adapted



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# Seminal serial crystallography experiments performed at XFELs





10<sup>10</sup> times more brilliance than 3<sup>rd</sup> generation synchrotron

Boutet et al., 2012, Science Redecke et al., 2012, Science Chapman et al., 2011, Nature

## **Principle behind XFELs**

Idea of FEL: J. M. J. Madey (1971) J.Appl. Phys.42, 19061



# Self-Amplified Spontaneous Emission (SASE) due to continuous interaction of

emitted X-rays with the electron bunch over the full undulator length



SASE in long undulator (100 m) leads to micro-bunching of electron macro-bunches

Micro-bunches are separated by a distance equal to one radiation wavelength:

#### **Coherent wave is emitted with very high brilliance**

# **Diffraction before desctruction imaging**



## Serial femtosecond crystallography (SFX):



Chapman et al., 2011, Nature

Boutet et al., 2012, Science;

Redecke et al., 2012, Science

## Main limitations of SFX: beamtime availibility and sample consumption

- 2 new XFELs under construction : European XFEL (2016) and SwissFEL (2017)

- Alternative approaches to the gas-focused liquid jet (14 mL per 12 hours):

- Lipid cubic phase injector: 1-200  $\mu$ L/12 hours, but LCP may affect diffraction
- Nanoflow electrospinning injector: 149  $\mu$ L/ 12 hours, but no salt...



Weierstall et al., (2014), Nat. Comm.



## Is serial crystallography amenable to synchrotron sources?

- *T. brucei* cathepsin B crystals (CatB) mounted in a nylon loop at 110 K (Gati et al, 2014, IUCR J)
- Lysozyme crystals mounted in a poly-vinyle chip (Zarrine-Afsar et al., 2012, Acta D)
- →Statistics of 'acceptable' quality
  →Oscillation data collection

## Is REAL serial crystallography amenable to synchrotron sources?

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→Statistics of 'acceptable' quality
→Oscillation data collection

- $\rightarrow$  We need high-quality data
- $\rightarrow$  We need an handling-free & centering-free methodology

# Crystals can be presented to the X-ray beam on solid support

# and data collected in raster-scanning mode (no oscillation)



We collect tens of thousands of diffraction patterns...

Sometimes, we hit a crystal...

Sometimes, we hit solvent...

We need to split the difference !!!

# Sorting the good from the bad ?

"Hit-finding" is a major challenge in serial crystallography

At XFELs, approaches have been developed (Cheetah, Apple.Py), yet they are not suited for conventional X-ray crystallography data formats (edf, cbf, smv, mccd, ...)

Also, these approaches do not include background subtraction nor background classification and scaling.



If it isn't available, do it yourself !!!



## NanoPeakCell : a hit finder that understands data need to be corrected...

000	NanoPeakCell	
Directory /windirect/_data_vial Browse Detailet %v4_cell1_1009	87	Viewer Settings Intensity Boost -0
Dependential Setup Distance (mm) 98.605 Wavelength (A) 9.832 Beam Center: X 123 X 136		Min Value 0 10 100 Max Value 100 1000 Cotour Magaring Bures 1
Instituter Parameters		Progression and results Job Progression
Threshold 50 Min number of pixels 20 Number of cpus to use: 12		Current Hit Rate
Image Convection and Output formatial Detector Correction (Ratheld-distantion) Kackground Substraction		Load results: Browse
Reckpround images: 1 Number of images for blig: 10		
Convert Ries in I		
add the format		
Find Hits T		
	♦ O O + S 🗄 😸 × v Intensity:	Pay Stop

### NanoPeakCell : a hit finder that prepare data for further processing...



# Proof of concept on lysozyme :

Parameter	"micro"		"nano"	
Resolution	50 - 1.7 Å		50 - 1.5 Å	
Beam size (µm (v) * µm (h))	1.5 * 2.5		0.150 * 0.175	
Beam divergence [mrad]	1.0		0.326	
Wavelength [Å]	0.954		0.832	
Beam fluence [ph/s]	1.00 x 10 <sup>11</sup>		1.70 x 10 <sup>10</sup>	
Flux density [ph/s/mm <sup>2</sup> ]	2.67 x 10 <sup>19</sup>		6.48 x 1017	
Brilliance [ph/s/mm <sup>2</sup> /mrad <sup>2</sup> ]	2.67 x 10"		6.09 x 10**	
Bandwidth [Å]	< 1.00 x 10 <sup>-4</sup>		< 1.00 x 10 <sup>-4</sup>	
Exposure time per frame [s]	0.2		0.1	
Dose [MGy] per crystal	2.70		24.72	
Dose rate [MGy/s]	27.0		240.7	
Space Group		P	4.2.2	
Indexing software	CrystFEL	cctbx.xfel	CrystFEL	cctbx.xfel
Unit cell length	a=b=78.0 Å	a=b=78.0 Å	a=b=78.0 Å	a=b=78.0 Å
[Å], α=β=γ=90°	c= 38.3	c= 38.4	c= 38.5	c= 38.4
# collected diffraction images	69,319		139,985	
# hits/indexed	6999/3576		45172/35446	
# reflections	5997182		3564462	
# unique reflections	13546		19613	
Completeness [%]	100.0 (100)		100.0 (100)	
Redundancy	442.7 (60.6)		133.3 (47.8)	
l/ol	5.58 (3.71)		11.23 (8.52)	
Rsplit	18.34 (32.30)		6.18 (21.28)	
Rsym (I)	25.28 (44.49)		8.70 (21.72)	
Rsym (F)	14.00 (22.82)		5.42 (10.81)	
CC <sup>1/2</sup> [%]	0.99 (0.81)		0.99 (0.54)	
Wilson B [Å <sup>2</sup> ]	21.71		11.07	
Rfree [%]	29.97		25.11	
Rfact [%]	23.82		21.52	
Rmsd bonds [Å]	0.007		0.008	
Rmsd angles [*]	1.040		1.040	

#### Table 1. SFX and synchrotron data and refinement statistics

Parameter	40 fs* pulses	5 fs* pulses	SLS RT data 3 ****
Wavelength	1.32 Å	1.32 Å	0.9997 Å
X-ray focus [µm <sup>2</sup> ]	~ 10	~ 10	$\sim 100 \times 100$
Pulse energy/fluence at sample	600 µJ/ 4x10 <sup>11</sup> ph/pulse	53 µJ/3.5x10 <sup>10</sup>	N.A./ 2.5 x10 <sup>10</sup> ph/s
Dose [MGy]	33.0 per crystal	2.9 per crystal	0.024 total
Dose rate [Gy/s]	$8.3 \times 10^{20}$	5.8 × $10^{20}$	9.6 × 10 <sup>2</sup>
Space group	$P4_{3}2_{1}2$	$P4_{3}2_{1}2$	$P4_{3}2_{1}2$
Unit cell length [Å], α=β=γ=90°	a=b=79, c=38	a=b=79, c=38	a=b=79.2, c=38.1
Oscillation range/exposure time	Still exp. / 40 fs*	Still exp. / 5 fs*	1.0°, 0.25 s
# collected diffraction images	1471615	1997712	100
# of hits/indexed images	66442 /12247	40115/10575	n.a./100
Number of reflections	n.a.	n.a.	70960
Number of unique reflections	9921	9743	9297
Resolution limits [A]	35.3-1.9	35.3-1.9	35.4-1.9
Completeness**	98.3% (96.6%)	98.2% (91.2%)	92.6% (95.1%)
I/ $\sigma$ (1)**	7.4 (2.8)	7.3 (3.1)	18.24 (5.3)
R <sub>split</sub> ***	0.158	0.159	n.a.
R <sub>merge</sub>	n.a.	n.a.	0.075 (0.332)
Wilson B-factor*****	28.3 Å <sup>2</sup>	28.5 Å <sup>2</sup>	19.4 Å <sup>2</sup>
R-factor/R-free*****	0.196/0.229	0.189/0.227	0.166/0.200
Rmsd bonds, Rmsd angles*****	0.006 Å, 1.00°	0.006 Å, 1.03°	0.007 Å, 1.05°
* Electron hungh length ** Lighast re-	4E18	4E19	4EIU

\*\* R<sub>solit</sub> as defined in (15) 
$$R = \left(\frac{1}{r}\right) \left(\sum_{kl} \left|I_{kll}^{even} - I_{kll}^{odd}\right| / \frac{1}{2} \sum_{kll} \left|I_{kll}^{even} + I_{kll}^{odd}\right|\right)$$

\*\*\* 
$$R_{split}$$
 as defined in (15)  $R_{gas} = \left(\frac{1}{\sqrt{2}}\right) \cdot \left(\sum_{hkl} \left|I_{kkl}^{even} - I_{hkl}\right| / \frac{1}{2} \sum_{hkl} \left|I_{kkl}^{even} + I_{hkl}^{ada}\right|\right)$ 

\*\*\*\* Statistics from XDS (19) \*\*\*\*\* Calculated with TRUNCATE (20) \*\*\*\*\*\* Calculated with PHENIX (21)

From Boutet et al., 2012, Science

- Proof of feasibility for serial synchrotron crystallography (SSX) using micro (1.5  $\mu m$ ) or nano (150 nm) focused beams

- Data of comparable or better quality than that produced by XFELs (...), yet using much less frames.

- An comparatively economic approach:  $\sim 5\mu L$  of concentrated crystals instead of tens of mL at XFELs.

### Can this approach be used to characterize naturally-occuring nano-crystals?



Nano-crystals of Cry3A in *Bacillus thuringiensis* cells

Nano-crystals of B1nAB in *Bacillus thuringiensis* cells

Nano-crystals of major basic protein in human eosinophilic granulocytes

# Macromolecular nano-crystallography at ESRF

#### 50\*50 to 200\*200 nm2 X-ray beam on ID13EH3

Manfred Burghammer & Christian Riekel



Long story short: impossibly small nanocrystals are not amenable to SSX.

# We need to resort to SFX at XFELs : the ultra-short nature of the pulses will allow observing diffraction before destruction occurs



XFELs deliver 10-10000 times more X-rays in 50 fs than at ESRF in 1s...

# Liquid jet consumes 14 ml of sample (10<sup>10</sup>-10<sup>11</sup> crystals/ml) per 12h ...





Feasible for lysozyme ...

# But not (humanely) feasible for naturally occuring human nanocrystals



Schlichting & Miao (2012) Curr Opin Struct Biol

NB: cells are purified from the blood of donors (patients with hyper-eosinophilic auto-immune diseases)

# Phasing of SFX data is notoriously difficult....

- •To date, only Schlichting and coll. have succeeded in phasing SFX data Barends et al, 2014, Nature
- •Amidst reasons :
  - Lack of isomorphism between crystals (>  $10,000 \neq$  crystals)
  - Energy jittering between XFEL pulses
  - Large bandwidth

Could be solved by self-seeding



Amann et al., 2012, Nat. Photonics Ding et al., 2010, Phys. Rev. ST Accel. Beams



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### X-ray scattering from a molecule



# Sample space

# **Scattering space**

<u>Diffraction Pattern</u>: contains all the contrast relevant information at the resolution of  $\lambda/2\sin(\theta)$ 

# Ideal case :



# Single particle imaging by coherent X-ray imaging : direct phasing

Essential: Direct phase retrieval by the oversampling technique (Miao et al. (2001) PNAS 98, 6641)



Essential: Methods from single particle EM

10<sup>7</sup> single shots necessary

#### Gaffney & Chapman (2007) Science 316, 1444

# Single particle imaging of a 500 nm particle : done !!!



Seibert et al., 2011, *Nature* Schlichting & Miao, 2012, *Curr Opin Struct Biol* 

### But protein are too small --- not enough scattering for direct phasing !!!

X-ray/matter interaction is weak (1/10000 photons)

We use crystals as a means to amplify the signal (2.10<sup>14</sup> symmetry-related molecules)

The Bragg law applies : constructive interferences only if  $2^*d^*sin(\theta) = \lambda$ 



transform; no Bragg spots

fringes between spots

# When using nanocrystals, convolution of the continuous fourier transform

# of the object with that of the lattice

e.g. Photosystem I (P6<sub>3</sub>)

- unit cell : 281\*281\*165 Å<sup>3</sup> α=β= 90°; γ=120°
- 8 Å wavelength, 1 µm<sup>2</sup> beamsize, 15,000 frames

#### N unit cells gives rise to N-2 fringes between neighbouring Bragg peaks.

(fringe-spacing is finer by a factor of 1/N than the Bragg spacing)



Information between bragg peaks could be used to phase the structural information



# Best use of XFEL for now : (time-resolved) micro-crystallography :

# e.g. Photosystem I (P6<sub>3</sub>)

- unit cell : 281\*281\*165 Å3
  - α=β= 90°; γ=120°
- 8 Å wavelength, 1  $\mu m^2$  beamsize, 15,000 frames

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# Information between bragg peaks could be used to phase structural information



Rodriguez & Miao, 2014, IUCR J.

# Some proteins form nano-crystals in vitro :

e.g. membrane proteins, amyloid proteins, protein complexes...

- Check out whitish precipitates in crystallization drops

Do TEM on your drops !!!



Nelson et al., 2005, *Nature* Sawaya et al., 2007, *Nature* 

- Keep faith... (0.8 Å data from a nano-crystal after 8h of "fish-mount-shoot" routine)



Nasrallah et al., submitted

- Refrain from avoidance of the feasible by invocation of the impossible

Shoot first, think later !!!

## Some proteins form nano-crystals in vivo:

- Opens a new era in structural biology, in which the **structures** of macromolecules are probed directly in their **cellular environments** 

- **XFEL** >>> synchrotron radiation because crystals are small

- In the near future: 4<sup>th</sup> generation synchroton sources : 10<sup>4</sup> increase in flux !!!

 $\rightarrow$  may become feasible at synchrotron too ?

# Serial crystallography :

- Already feasible at synchroton sources (use NanoPeakCell !!!)

- Serial crystallography is important for macromolecular objects that are hard to crystallize :

   \* membrane proteins
   \* protein complexes
- Time-resolved crystallography : pump-and-probe experiments on the fs-ms timescale

# In conclusion :

#### - It's a great time to be doing structural biology !!!!

(mass spec., spectroscopists, crystallographers, programmers, biologists, engineers...)

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